



Simulations and Analyses of Li₄ and Li₆ Boracites and Thioboracites as Promising Li Ion Conducting Electrolytes

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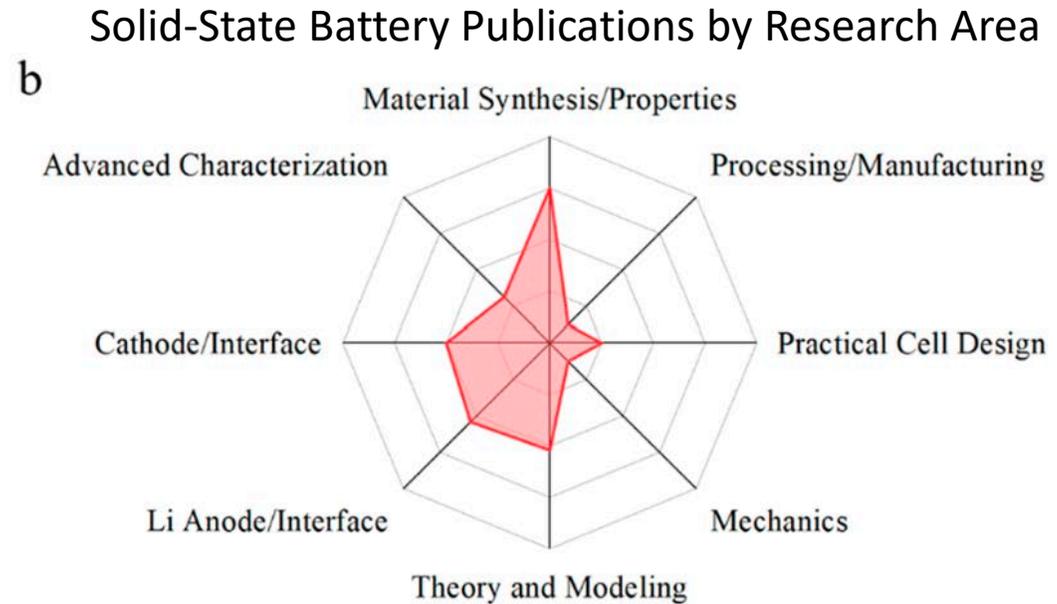
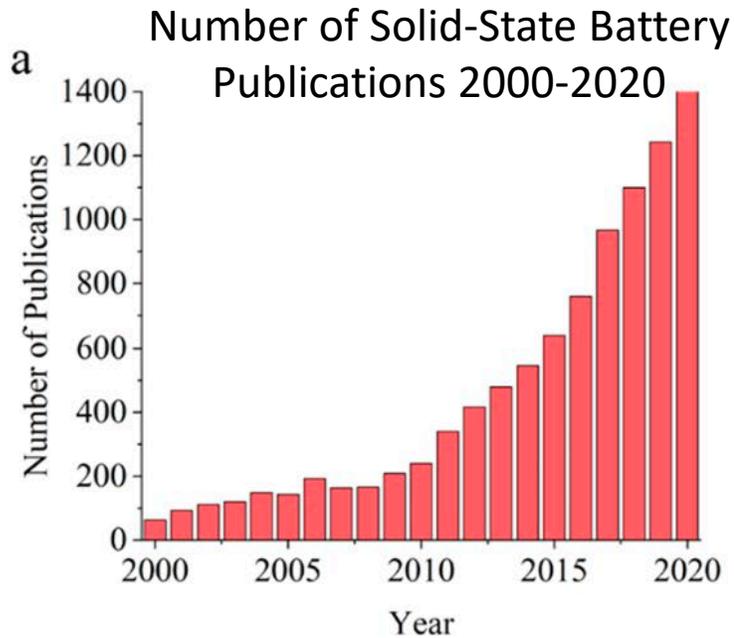
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Grant: DMR-1940324

Research Motivation: Active Field Interest



a.) Number of peer-reviewed scientific publications relating to solid-state lithium battery research from 2000-2020

b.) Solid-state lithium battery research breakdown by type

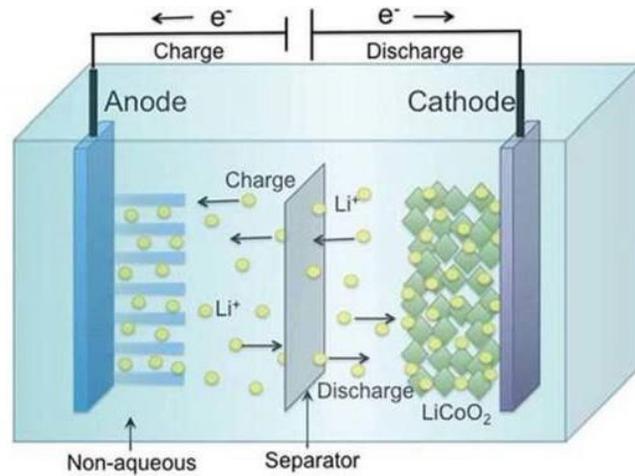
(Image credit: ACS Energy Letters, 2021 DOI: 10.1021/acseenergylett.1c00445)

- Publications in solid-state battery research increased by a factor of ~ 7 from 2010-2020
- Significant research interest is focused on Material Synthesis/Properties
- This leaves Theory and Modelling as an area in need of more research

Research Motivation: Why Solid-State Batteries?

a.)

Li-ion Battery (Liquid Electrolyte)



a.) Schematic illustration of the internal components of a typical Li-ion battery.

(Image credit: Journal of The Electrochemical Society, 2017. DOI: 10.1149/2.1451709jes)

b.) Schematic illustration of an ideal high-energy Solid-State battery stack.

(Image credit: ACS Energy Letters, 2021. DOI: 10.1021/acseenergylett.1c00445)

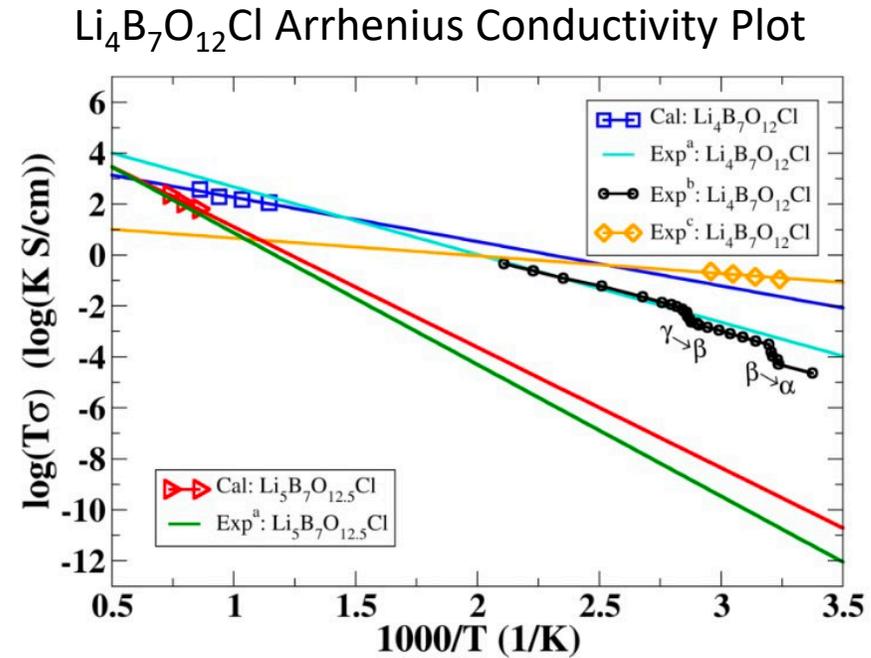
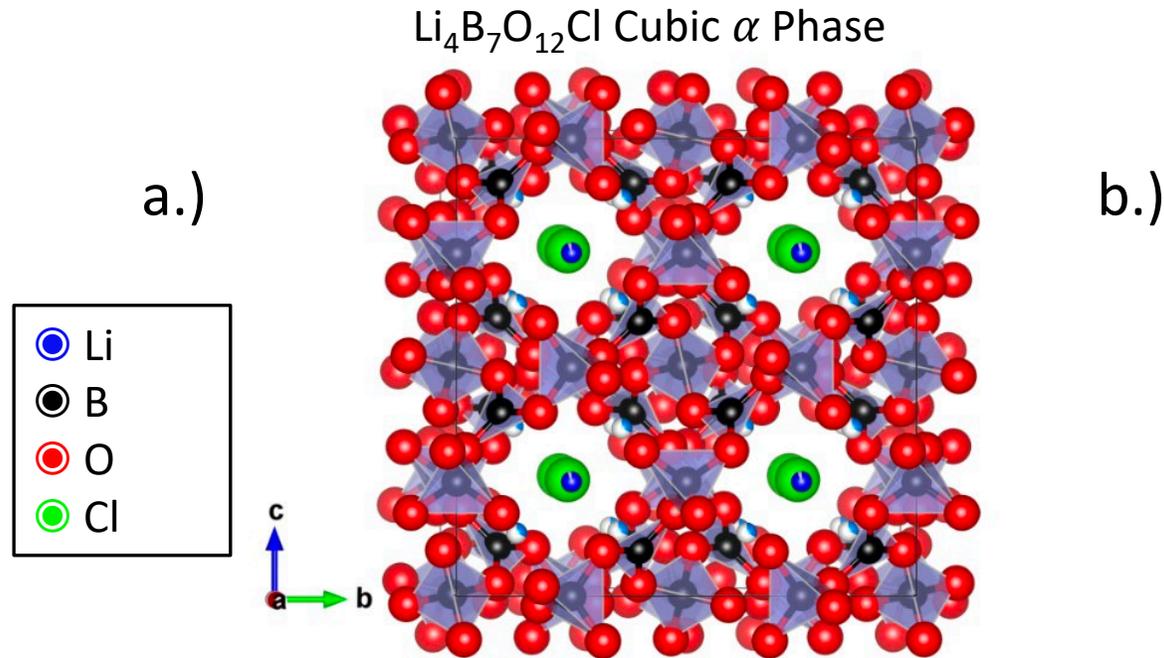
Solid-State Li-ion Battery Stack (Solid Electrolyte)

b.)



- Pure lithium anode can greatly improve performance
- Solid electrolytes have the following potential benefits:
 - Expected to be stable in contact with pure lithium
 - Competitive ionic conductivity with liquid electrolytes and negligible electronic conductivity
 - Improved safety by addressing leakage of toxic liquid electrolytes
 - Improved safety by addressing fire/explosion risk from flammable liquid electrolytes
 - Improved longevity by replacing liquid electrolytes that decompose and reduce functionality

Research Motivation: Previous Boracite Results

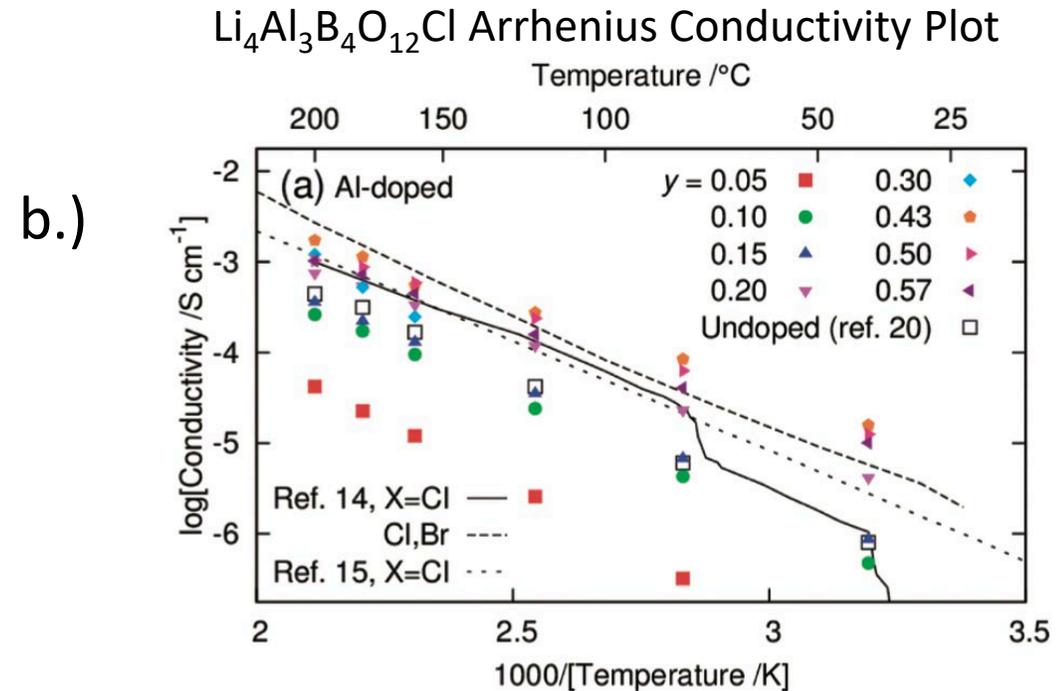
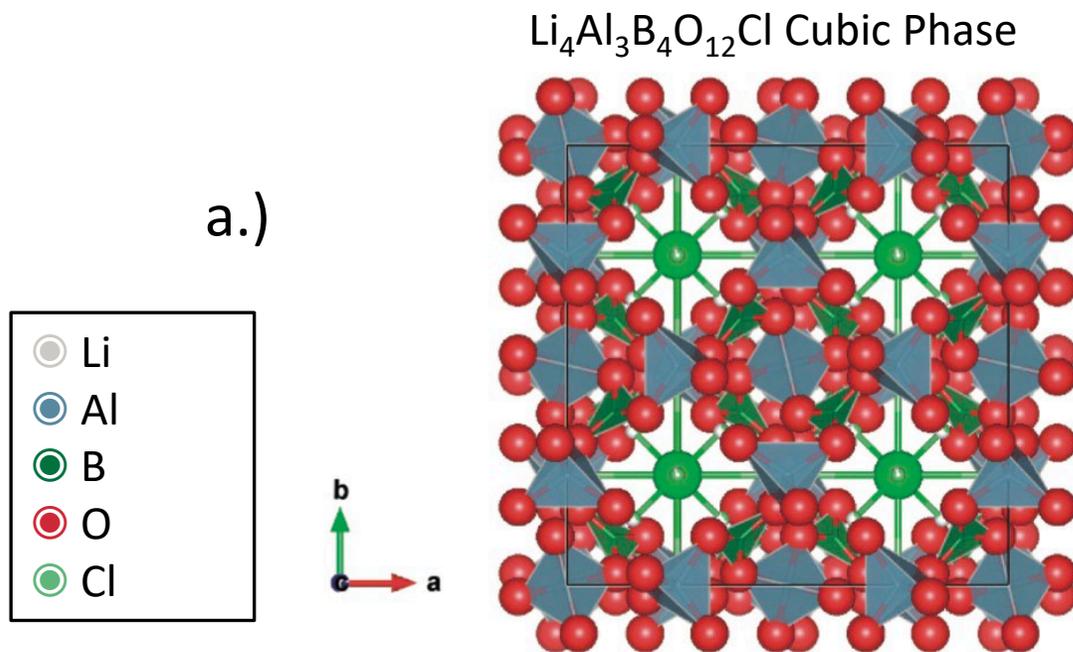


a.) $\text{Li}_4\text{B}_7\text{O}_{12}\text{Cl}$ experimental structure and b.) Arrhenius plot of the conductivity vs. temperature

Image credit: Li and Holzwarth, 2022. DOI: 10.1103/PhysRevMaterials.6.025401

- Face Centered Cubic ($F\bar{4}3c$) α -phase reported at room temperature
- Disordered Lithium sites: 25% fractional occupancy
- Conductivity reported to be on the order of $3 \times 10^{-4} \text{ S/cm}$ at room temperature by Tan et. al. (DOI: 10.1021/acsaem.9b00812)
- Work by Dr. Li and Dr. Holzwarth agree with experimental performance results with new discoveries:
 - G.S. is Face Centered Rhombohedral ($R3c$) distortion of the cubic phase
 - 25% fractional Li sites in $F\bar{4}3c$ map to fully occupied Li sites in $R3c$, remaining 75% map to interstitial sites
 - Ion migration mechanisms involving concerted ion motion and interstitial sites

Research Motivation: Partial Al Substitution



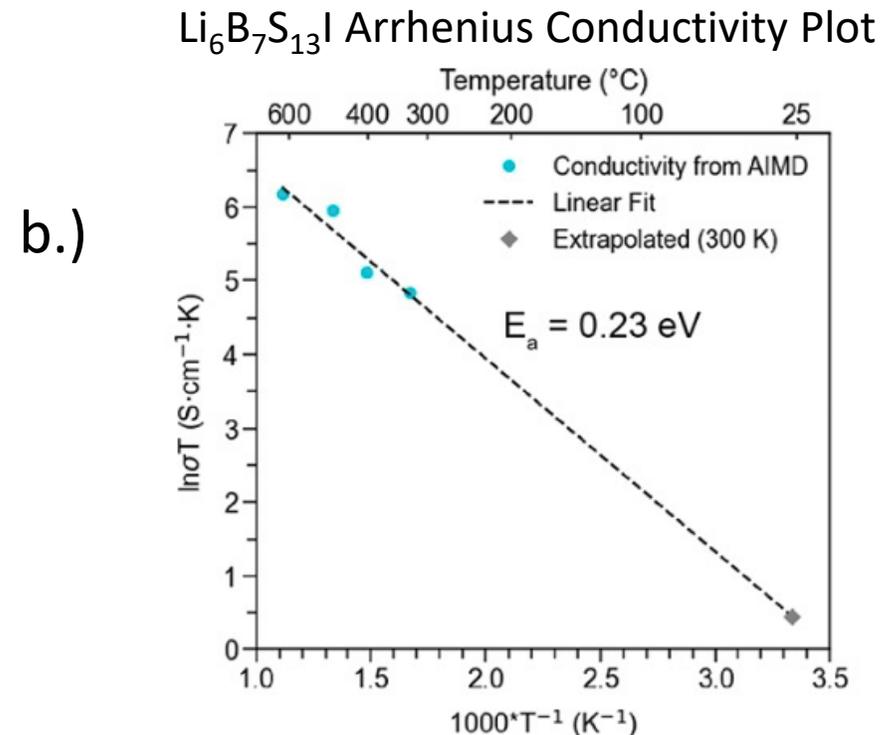
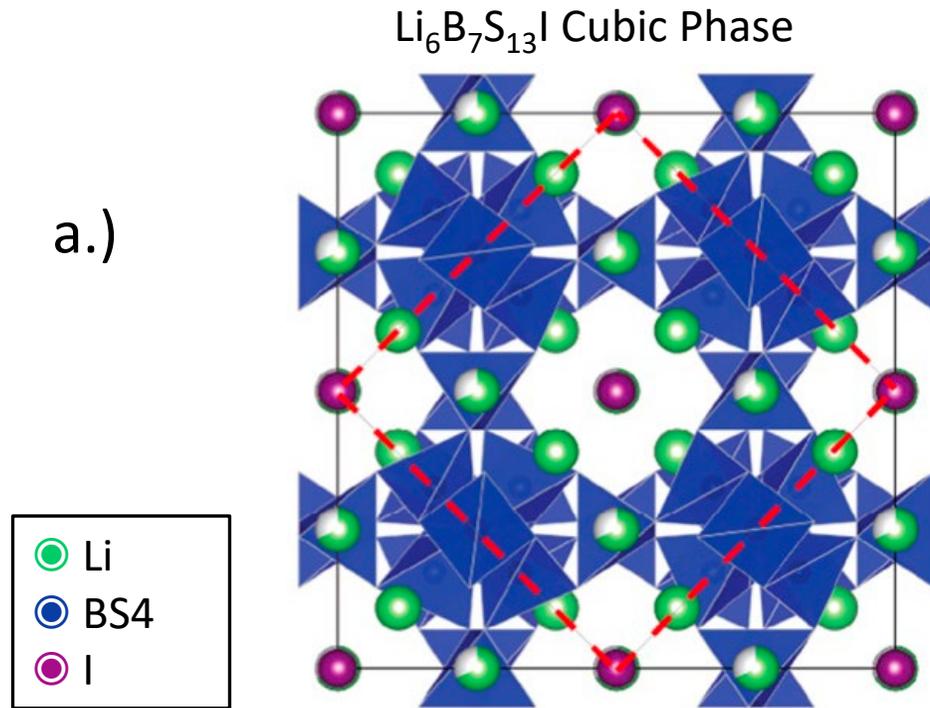
$\text{Li}_4\text{Al}_3\text{B}_4\text{O}_{12}\text{Cl}$ experimental structure (a) and Arrhenius plot of the conductivity vs. temperature (b)

Image credit: Kajihara et. al., 2017.

DOI: 10.1246/bcsj.20170242

- Experimental investigation of substituting boron sites in $\text{Li}_4\text{B}_7\text{O}_{12}\text{Cl}$ with aluminum
- Found optimal y value of 0.43 : the ratio of Al_2O_3 to B_2O_3 in the starting synthesis mixture
- Synthesized $\text{Li}_4\text{Al}_3\text{B}_4\text{O}_{12}\text{Cl}$ and determined some of its properties:
 - Room temperature structure is similar to the original material: Face Centered Cubic ($F\bar{4}3c$)
 - Has similar performance - conductivity reported to be on the order of $1 \times 10^{-5} \text{ S/cm}$
 - Reported to be stable in contact with pure lithium metal

Research Motivation: S Substitution & Increased Li



$\text{Li}_6\text{B}_7\text{S}_{13}\text{I}$ experimental structure (a) and Arrhenius plot of the conductivity vs. temperature (b)

Image credit: Kaup et. al., 2021. DOI: 10.1021/jacs.1c00941

- Experimental investigation of similar material with sulfur and higher lithium stoichiometry
- Compound has iodine instead of chlorine as the halogen in the structure
- Similar room temperature Face Centered Cubic ($F\bar{4}3c$) phase found, as well as new tetragonal ($I4_1/a$) phase
- Face Centered Cubic ($F\bar{4}3c$) phase found to be stable, tetragonal ($I4_1/a$) phase reported to be metastable
- Material found to have high conductivity and stability
- Experimental conductivity calculated for cubic phase; found to be on the order of $5 \times 10^{-4} \text{ S/cm}$



In This Work

- Starting from known boracites and thioboracites:
 - $\text{Li}_4\text{B}_7\text{O}_{12}\text{Cl}$
 - $\text{Li}_4\text{Al}_3\text{B}_4\text{O}_{12}\text{Cl}$
 - $\text{Li}_6\text{B}_7\text{S}_{13}\text{I}$
- Investigating predicted boracites and thioboracites:
 - Sulfur substitutions of known $\text{Li}_4\text{-O}$ materials:
 - $\text{Li}_4\text{B}_7\text{S}_{12}\text{Cl}$
 - $\text{Li}_4\text{Al}_3\text{B}_4\text{S}_{12}\text{Cl}$
 - Chlorine substitution of known $\text{Li}_6\text{-S}$ material and similar derivatives (O, Al):
 - $\text{Li}_6\text{B}_7\text{S}_{13}\text{Cl}$
 - $\text{Li}_6\text{B}_7\text{O}_{13}\text{Cl}$
 - $\text{Li}_6\text{Al}_3\text{B}_4\text{S}_{13}\text{Cl}$
 - $\text{Li}_6\text{Al}_3\text{B}_4\text{O}_{13}\text{Cl}$

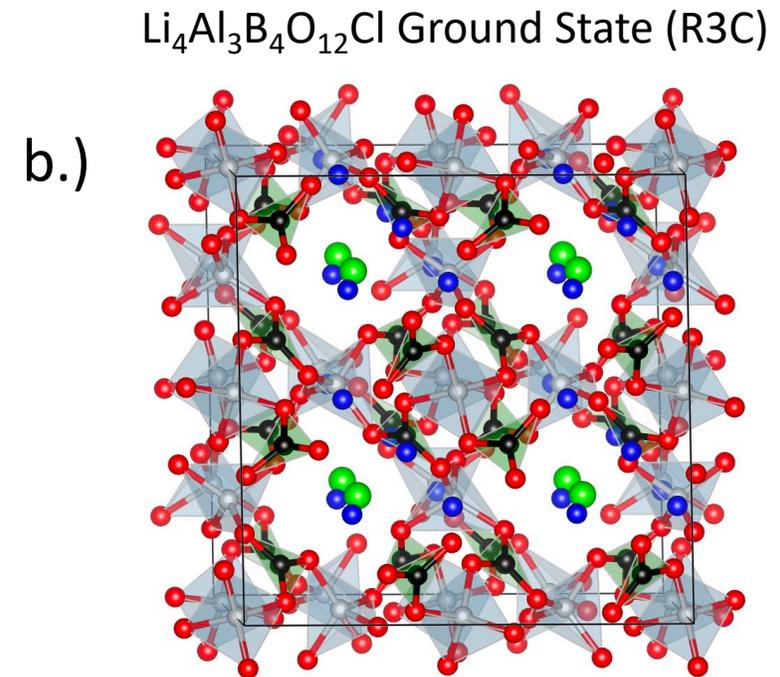
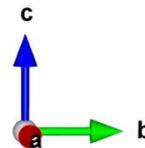
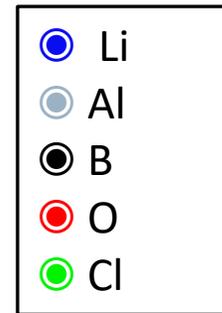
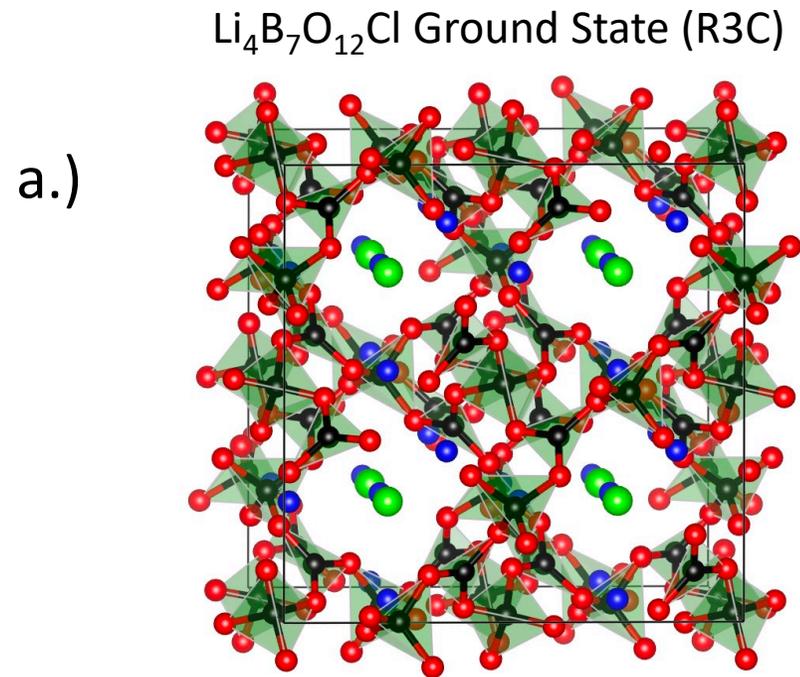


Computational Methods

- Ground State Structural Calculations
 - Density Functional Theory
- Phonon Density of States Calculations
 - Density Functional Perturbation Theory
- Ionic Diffusion and Conductivity
 - Ab Initio Molecular Dynamics
 - Born-Oppenheimer Approximation



Known Structures: $\text{Li}_4\text{B}_7\text{O}_{12}\text{Cl}$ and $\text{Li}_4\text{Al}_3\text{B}_4\text{O}_{12}\text{Cl}$



$\text{Li}_4\text{B}_7\text{O}_{12}\text{Cl}$ (a) and $\text{Li}_4\text{Al}_3\text{B}_4\text{O}_{12}\text{Cl}$ (b) rhombohedral R3c ground state structures

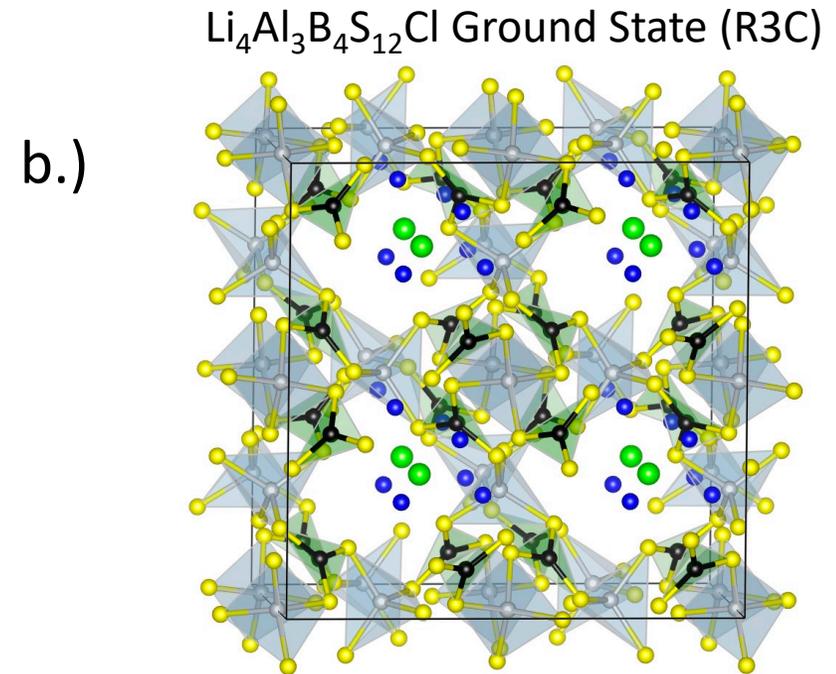
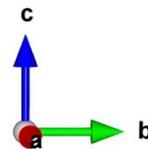
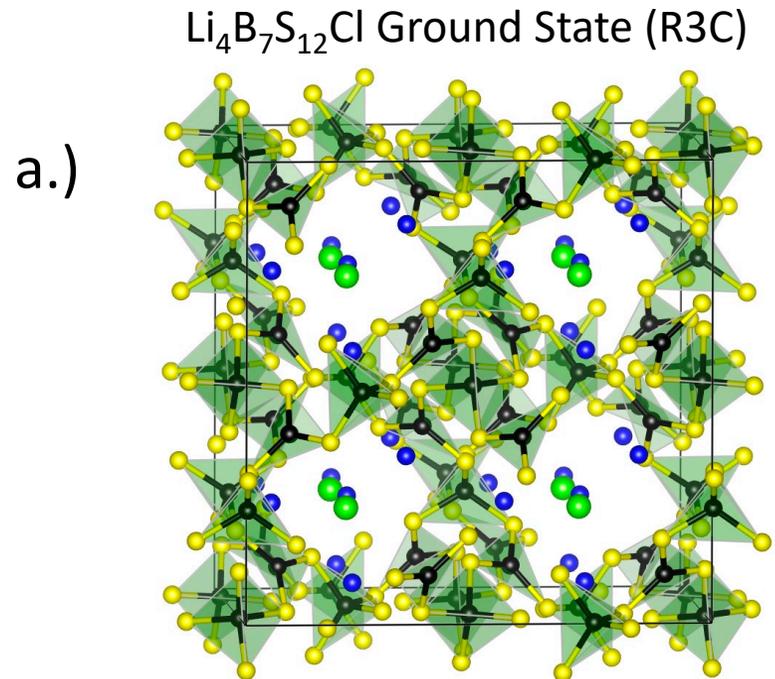
$\text{Li}_4\text{B}_7\text{O}_{12}\text{Cl}$ Conventional Cell

- R3c symmetry
- Lattice Constant = 12.1 angstroms
- Lattice Angle = 90.1 degrees

$\text{Li}_4\text{Al}_3\text{B}_4\text{O}_{12}\text{Cl}$ Conventional Cell

- R3c symmetry
- Lattice Constant = 13.0 angstroms
- Lattice Angle = 91.1 degrees

Predicted Structures: $\text{Li}_4\text{B}_7\text{S}_{12}\text{Cl}$ and $\text{Li}_4\text{Al}_3\text{B}_4\text{S}_{12}\text{Cl}$



$\text{Li}_4\text{B}_4\text{S}_{12}\text{Cl}$ (a) and $\text{Li}_4\text{Al}_3\text{B}_4\text{S}_{12}\text{Cl}$ (b) rhombohedral R3c ground state structures

$\text{Li}_4\text{B}_7\text{S}_{12}\text{Cl}$ Conventional Cell

- R3c symmetry
- Lattice Constant = 14.9 angstroms
- Lattice Angle = 89.8 degrees

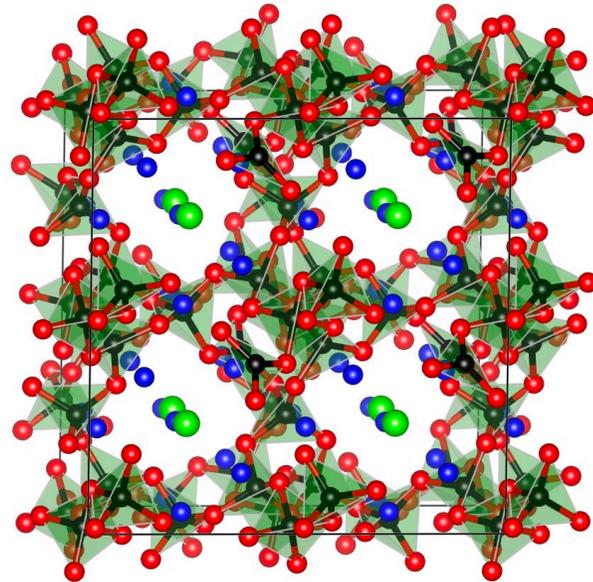
$\text{Li}_4\text{Al}_3\text{B}_4\text{S}_{12}\text{Cl}$ Conventional Cell

- R3c symmetry
- Lattice Constant = 16.1 angstroms
- Lattice Angle = 89.6 degrees

Predicted Structures: $\text{Li}_6\text{B}_7\text{O}_{13}\text{Cl}$ and $\text{Li}_6\text{Al}_3\text{B}_4\text{O}_{13}\text{Cl}$

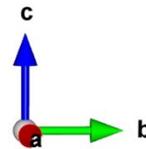
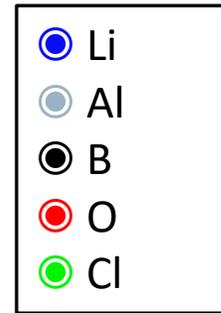
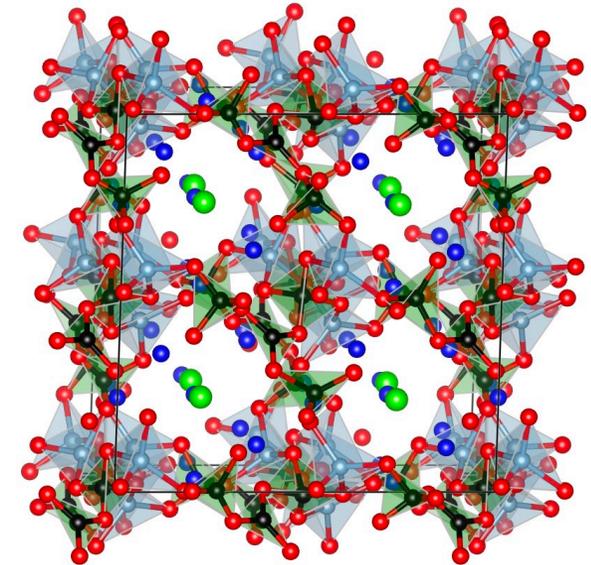
$\text{Li}_6\text{B}_7\text{O}_{13}\text{Cl}$ Ground State (R3C)

a.)



$\text{Li}_6\text{Al}_3\text{B}_4\text{O}_{13}\text{Cl}$ Ground State (R3C)

b.)



$\text{Li}_6\text{B}_7\text{O}_{13}\text{Cl}$ (a) and $\text{Li}_6\text{Al}_3\text{B}_4\text{O}_{13}\text{Cl}$ (b) rhombohedral R3c ground state structures

$\text{Li}_6\text{B}_7\text{O}_{13}\text{Cl}$ Conventional Cell

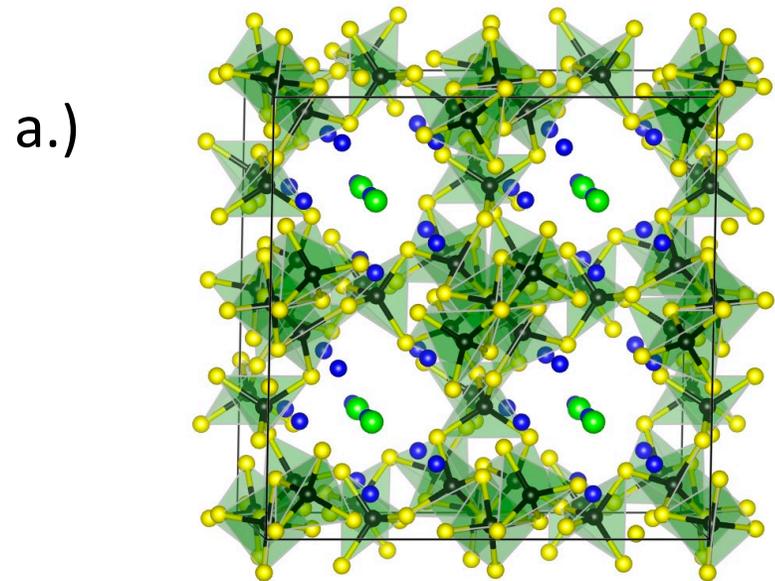
- R3c symmetry
- Lattice Constant = 12.2 angstroms
- Lattice Angle = 89.4 degrees

$\text{Li}_6\text{Al}_3\text{B}_4\text{O}_{13}\text{Cl}$ Conventional Cell

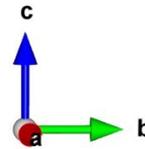
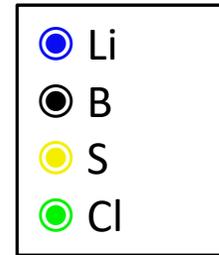
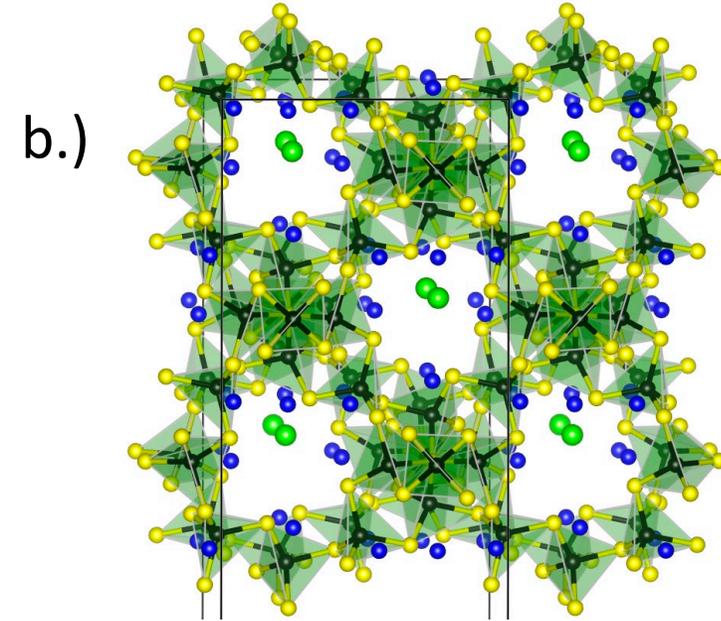
- R3c symmetry
- Lattice Constant = 13.0 angstroms
- Lattice Angle = 91.5 degrees

Predicted Structures: $\text{Li}_6\text{B}_7\text{S}_{13}\text{Cl}$ [R3c and Cc]

$\text{Li}_6\text{B}_7\text{S}_{13}\text{Cl}$ Rhombohedral Phase



$\text{Li}_6\text{B}_7\text{S}_{13}\text{Cl}$ Monoclinic Supercell



$\text{Li}_6\text{B}_7\text{S}_{13}\text{Cl}$ rhombohedral R3c (a) and monoclinic Cc (b) structures

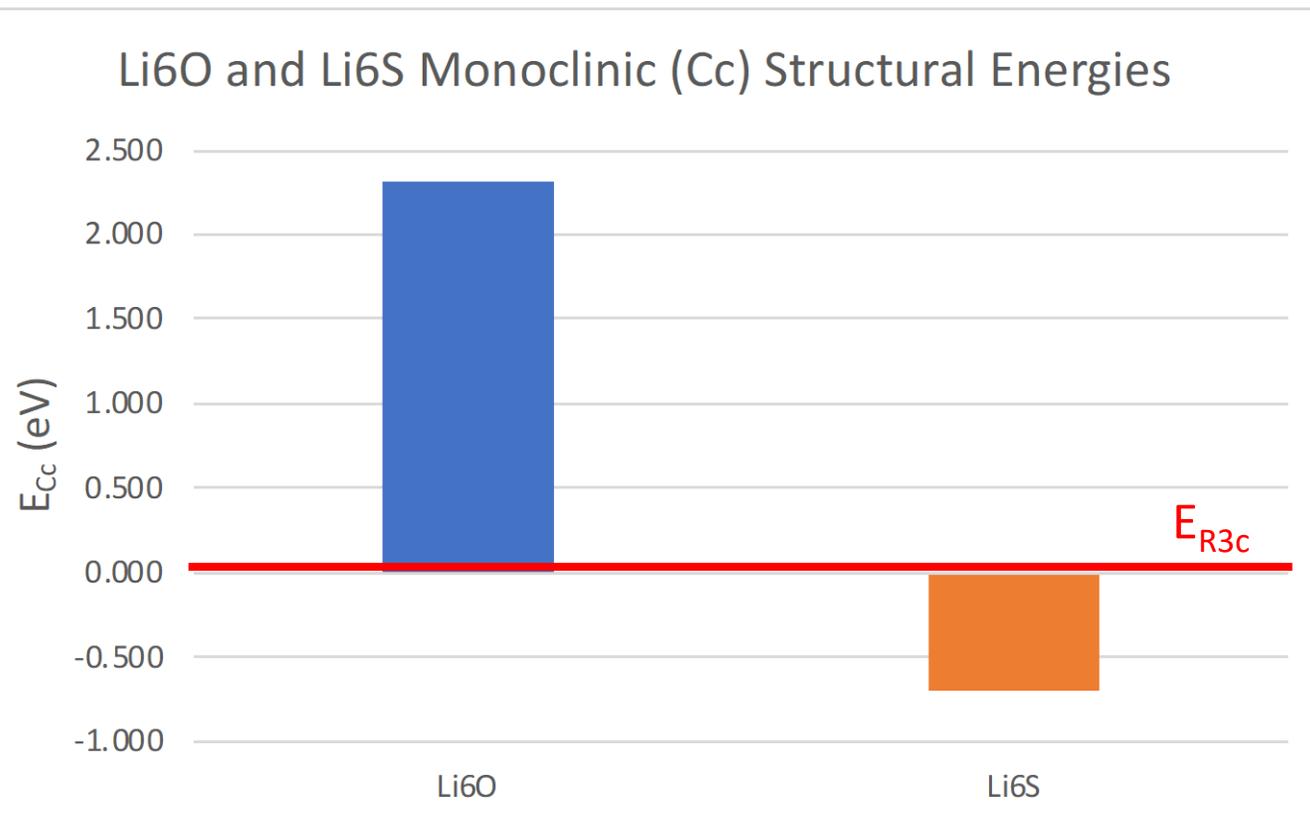
$\text{Li}_6\text{B}_7\text{S}_{13}\text{Cl}$ Conventional Cell

- R3c symmetry
- Lattice Constant = 15.1 angstroms
- Lattice Angle = 89.2 degrees

$\text{Li}_6\text{B}_7\text{S}_{13}\text{Cl}$ Conventional Cell

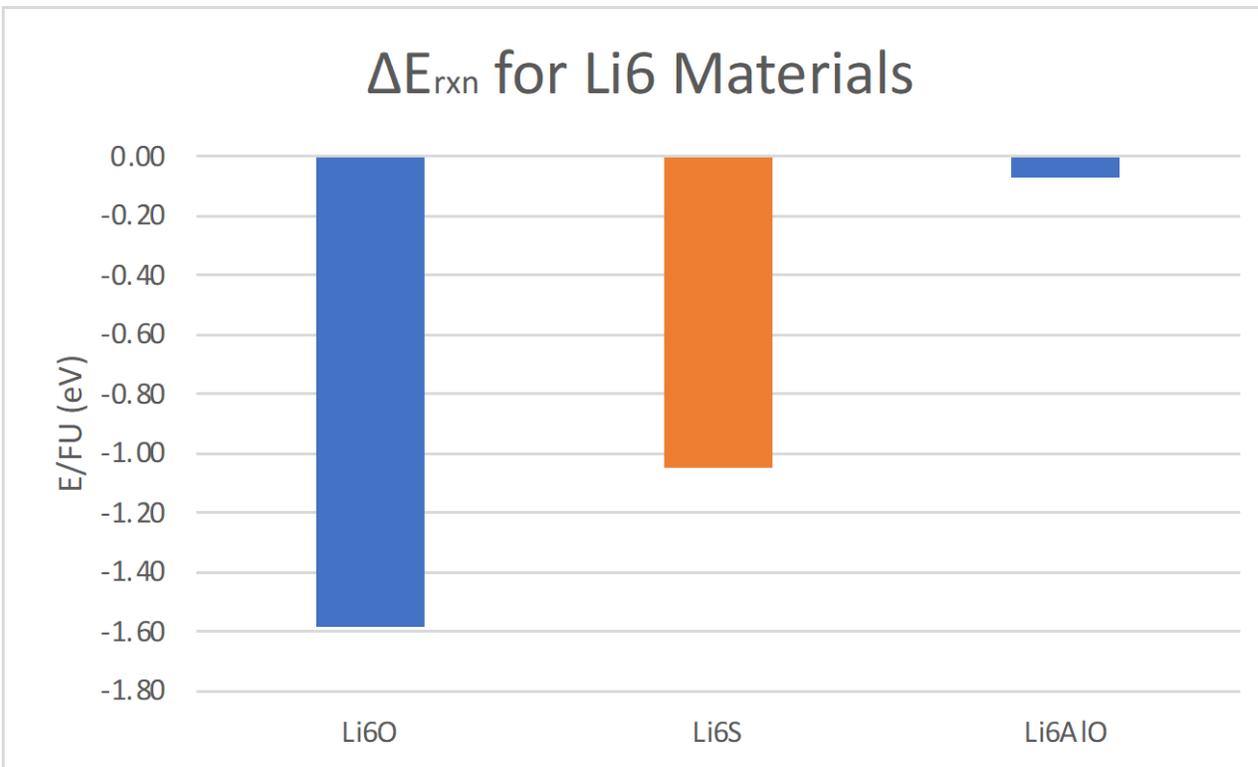
- Cc symmetry
- Lattice Constants = (18.5, 10.5, 10.8) angstroms
- Lattice Angles = (90.0, 124.6, 90.0) degrees

Finding the Ground State of the Li6 Materials



- Zero energy is set to the respective rhombohedral (R3c) structural energy
- For **Li6O**, we see the monoclinic structure is higher in energy by 2.31 eV
- This confirms the ground state structure is R3c as expected
- For **Li6S**, we see that the monoclinic structure is lower in energy by 0.70 eV
- This suggests the R3c structure is NOT the ground state structure
- The monoclinic structure is the current ground state candidate for Li6S

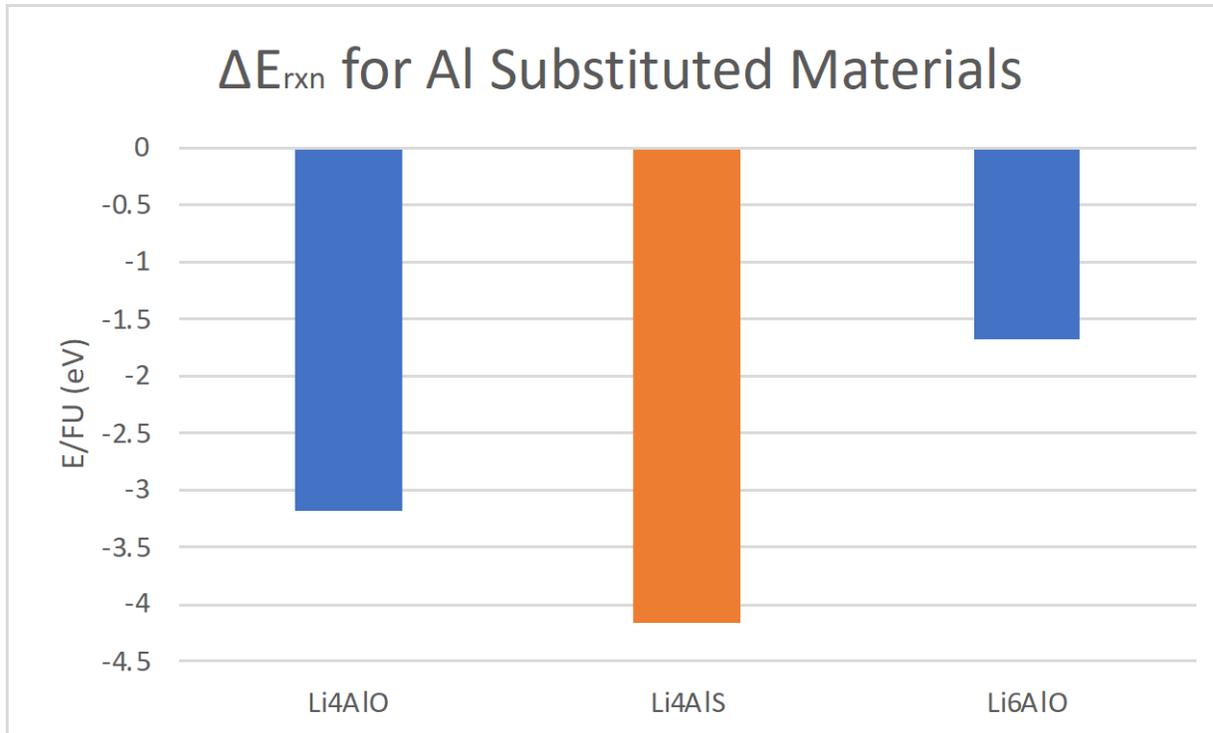
Chemical Stability: Li4 vs Li6



- ΔE_{rxn} Li6O: -1.58 eV / FU
- ΔE_{rxn} Li6S: -1.05 eV / FU
- ΔE_{rxn} Li6AlO: -0.07 eV / FU

- How stable are the Li6 materials compared to the Li4 materials?
- Look at reaction energy difference, ΔE_{rxn} , of decomposition reaction:
- $Li_6B_7X_{13}Cl \rightarrow Li_4B_7X_{12}Cl + Li_2X$
- X = O or S
- If $\Delta E_{rxn} = E_{left} - E_{right} < 0$, then the material is more stable than the decomposition products
- All **Li6 materials** are more energetically favorable than Li4

Chemical Stability: Al vs Pure Boron

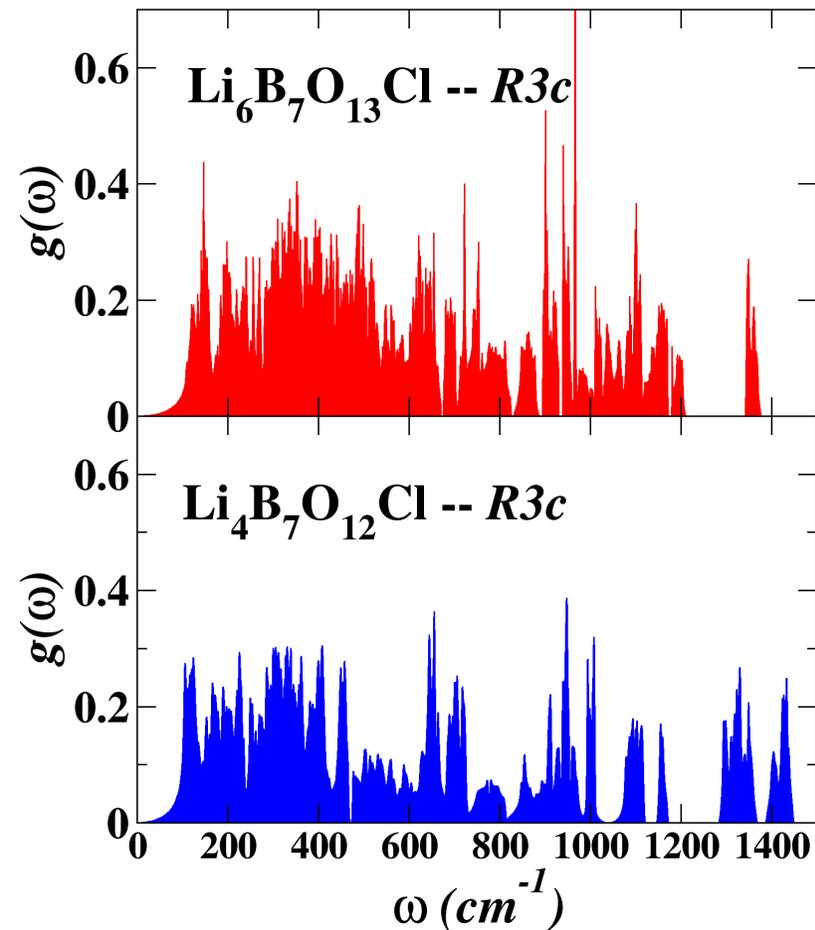


- ΔE_{rxn} Li4AlO: -3.18 eV / FU
- ΔE_{rxn} Li4AlS: -4.17 eV / FU
- ΔE_{rxn} Li6AlO: -1.67 eV / FU

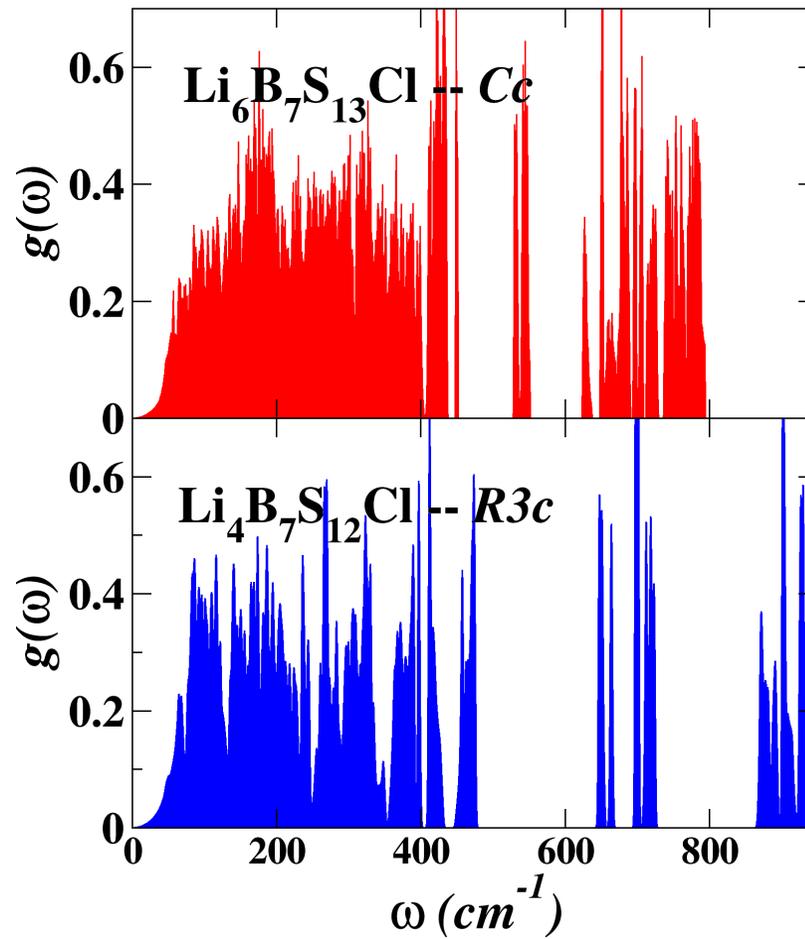
- How stable are Al substituted materials compared to pure boron materials?
- Look at reaction energy difference, ΔE_{rxn} , of decomposition reaction:
- $Li_n Al_3 B_4 X_m Cl + 3B \rightarrow Li_n B_7 X_m Cl + 3Al$
- X = O or S, n = 4 or 6, m = 12 or 13
- If $\Delta E_{rxn} = E_{left} - E_{right} < 0$, then the material is more stable than the decomposition products
- All **Al materials** are more energetically favorable than their pure boracites

Dynamic Stability: Phonon Density of States

Phonon D.o.S. for Oxygen Materials



Phonon D.o.S. for Sulfur Materials

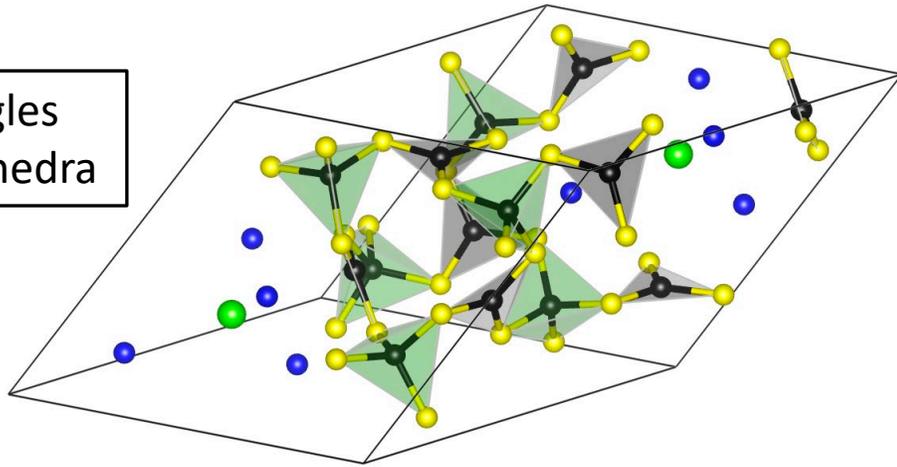
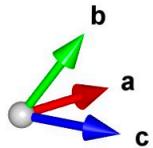


- $\text{Li}_6\text{S } R3c$ (not shown) has imaginary modes $\Rightarrow R3c$ not dynamically stable
- $\text{Li}_6\text{S } Cc$ is dynamically stable
- Li_4O , Li_4S , and $\text{Li}_6\text{O } R3c$ are dynamically stable
- Low frequencies $< 400 \text{ cm}^{-1}$ involve Li vibration
- Higher frequencies involve the boracite framework
- Highest frequencies are the triangular BO_3 modes

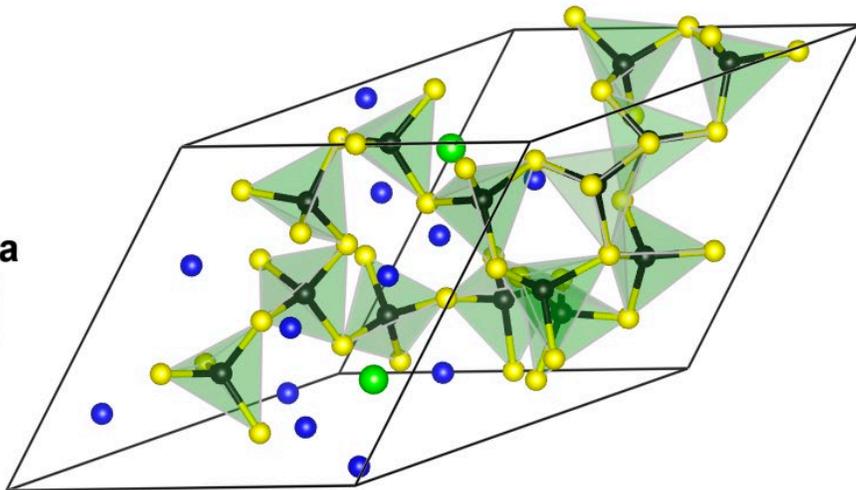
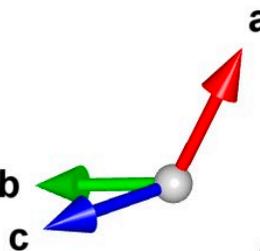
Dynamic Stability: Framework Geometry

$\text{Li}_4\text{B}_7\text{S}_{12}\text{Cl}$ Primitive Cell (R3c)

- BS3 triangles
- BS4 tetrahedra



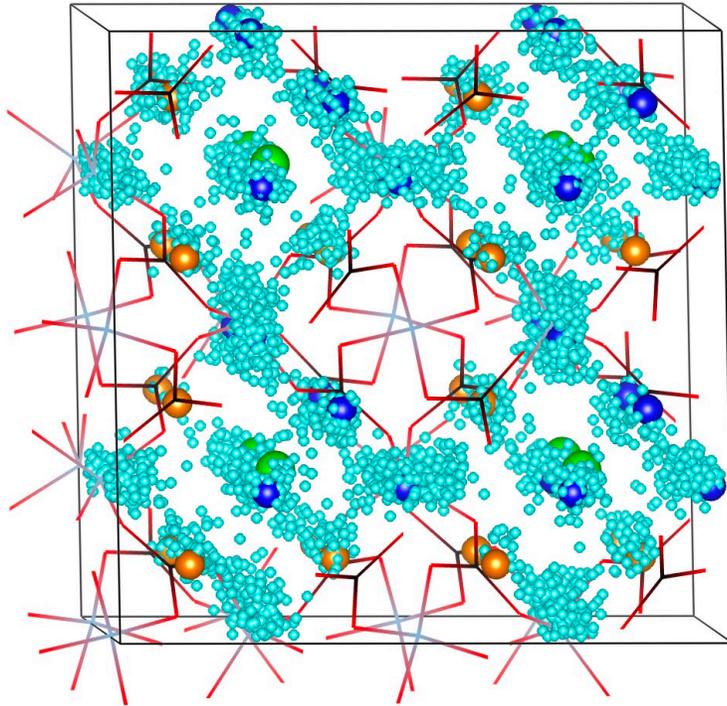
$\text{Li}_6\text{B}_7\text{S}_{13}\text{Cl}$ Primitive Cell (Cc)



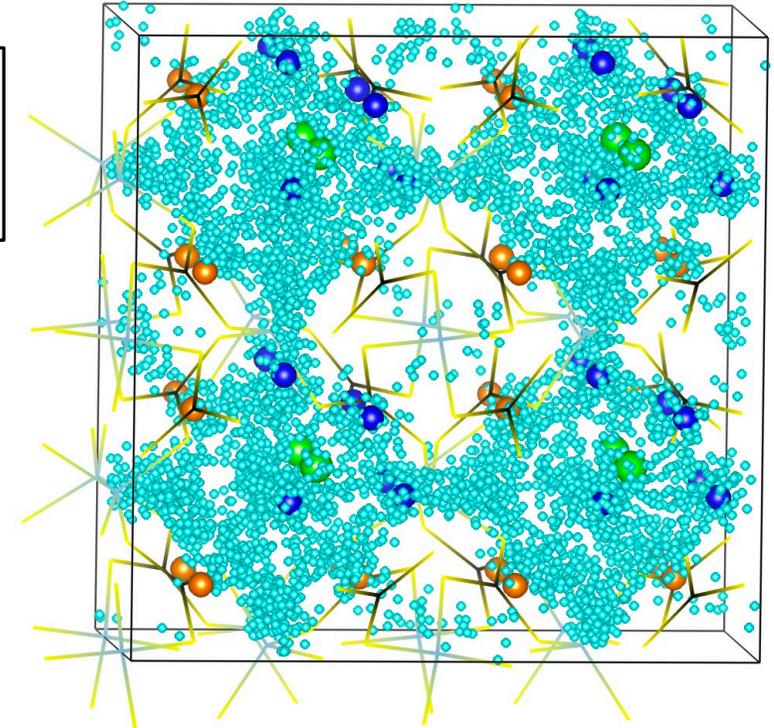
- Framework consists of BO_3/BS_3 planar triangles and BO_4/BS_4 tetrahedra
- Primitive Cell Analysis:
 - All Li_4 Structures:
 - 8 triangles (All B) and 6 tetrahedra (B or Al)
 - All Li_6O Structures:
 - 2 triangles (All B) and 12 tetrahedra (B or Al)
 - Li_6S Structures:
 - 14 tetrahedra, no triangles
- Structures with fewer BO_3/BS_3 triangles in the framework are more stable

Qualitative Performance: Sulfur vs. Oxygen

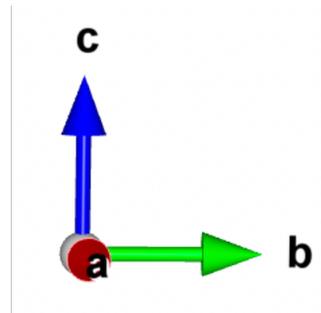
$\text{Li}_4\text{Al}_3\text{B}_4\text{O}_{12}\text{Cl}$ Li-ion Superposition



$\text{Li}_4\text{Al}_3\text{B}_4\text{S}_{12}\text{Cl}$ Li-ion Superposition



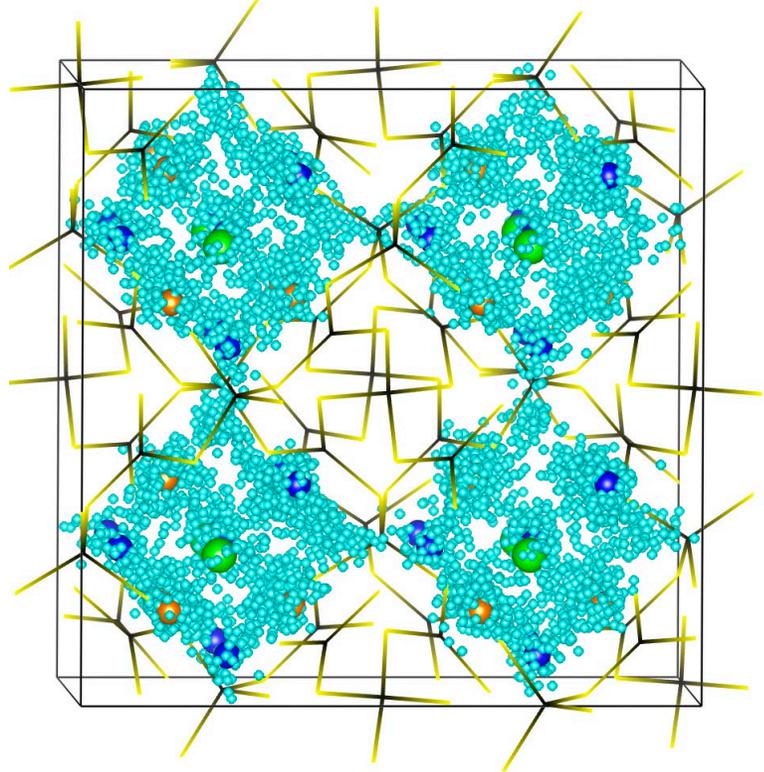
- Occupied Li Site
- Empty Li Interstitial Site
- Superimposed Li Positions



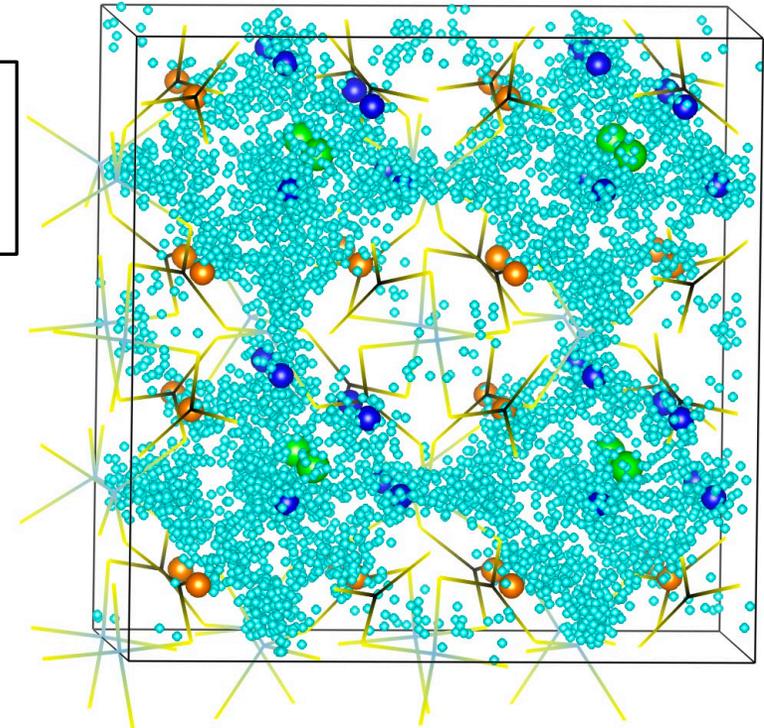
- $\text{Li}_4\text{Al}_3\text{B}_4\text{O}_{12}\text{Cl}$ (left) vs. $\text{Li}_4\text{Al}_3\text{B}_4\text{S}_{12}\text{Cl}$ (right) ion superposition diagrams
- Li positions superimposed over ~ 23 ps of simulation time
- Sampled every 50 timesteps at 1100K simulation temperature
- Much greater diffusion suggests sulfur significantly improves performance compared to oxygen

Qualitative Performance: Al Substitution vs. Pure B

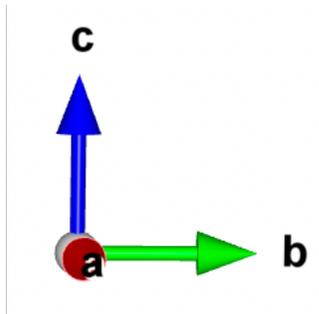
$\text{Li}_4\text{B}_7\text{S}_{12}\text{Cl}$ Li-ion Superposition



$\text{Li}_4\text{Al}_3\text{B}_4\text{S}_{12}\text{Cl}$ Li-ion Superposition



- Occupied Li Site
- Empty Li Interstitial Site
- Superimposed Li Positions



- $\text{Li}_4\text{B}_7\text{S}_{12}\text{Cl}$ (left) vs. $\text{Li}_4\text{Al}_3\text{B}_4\text{S}_{12}\text{Cl}$ (right) ion superposition diagrams
- Li positions superimposed over ~ 23 ps of simulation time
- Sampled every 50 timesteps at 1100K simulation temperature
- Greater diffusion may suggest Aluminum substitution improves performance compared to pure boron as well



Preliminary Summary

- Determined the ground state structures for predicted materials
 - All Li₄ materials, Li₆B₇O₁₃Cl, and Li₆Al₃B₄O₁₃Cl have rhombohedral (R3c) ground states
 - Li₆B₇S₁₃Cl has a newly discovered monoclinic (Cc) ground state structure
- Determined relative chemical stability between substitutions
 - Li₆ is energetically favorable compared to Li₄
 - Partial aluminum substitution is energetically favorable compared to pure boron
- Determined dynamic stability of predicted materials
 - All pure boron Li₄ and Li₆ materials are dynamically stable
 - R3c phase of Li₆B₇S₁₃Cl is not dynamically stable; ground state Cc is dynamically stable
- Determined qualitative measure of performance
 - Sulfur substitutions greatly improve diffusion and by extension, performance
 - Aluminum partial substitutions may also offer performance improvements



Future Investigation

- Find $\text{Li}_6\text{Al}_3\text{B}_4\text{S}_{13}\text{Cl}$ ground state
- Finish stability analysis on predicted materials
- Determine conductivities quantitatively with AIMD
- Investigate ion diffusion and ion migration mechanisms
- Investigate interface with Li metal