# Structure and interface properties of the electrolyte material Li<sub>4</sub>P<sub>2</sub>S<sub>6</sub>\*

## Zachary D. Hood,<sup>a</sup> Cameron M. Kates,<sup>b,c</sup> and <u>N. A. W. Holzwarth</u><sup>b</sup>

<sup>a</sup>Center for Nanophase Materials Sciences, Oak Ridge National Laboratory, Oak Ridge, TN 37831, <sup>b</sup>Department of Physics, Wake Forest University, Winston-Salem, NC, USA, 27109, <sup>c</sup>Currently attending the Pratt School of Engineering at Duke University

\*Supported by NSF Grant DMR-1105485. Computations were performed on WFU's DEAC cluster. ZDH was supported by the HERE program at ORNL. Helpful discussion with Chengdu Liang, Gayatri Sahu, Hui Wang, Melanie Kirkham, Jong Keum, and E. Andrew Payzant at ORNL and William C. Kerr, Keerthi Senevirathne, Cynthia Day, and Abdessadek Lachgar at WFU are gratefully acknowledged.



2015 APS March Meeting

## Li<sub>4</sub>P<sub>2</sub>S<sub>6</sub>

- Why-- Part of search for ideal solid electrolyte materials for allsolid state Li ion batteries
  - Reported in by Mercier et. al. , J. Solid State Chemistry 43, 151-164 (1982); hexagonal structure with disorder on the P sites
  - Frequently identified as unintended constituent of solid electrolyte preparations; relatively stability in air
- What-- Combined experimental and computation study including:
  - Structural analysis
  - Thermal stability
  - Transport properties



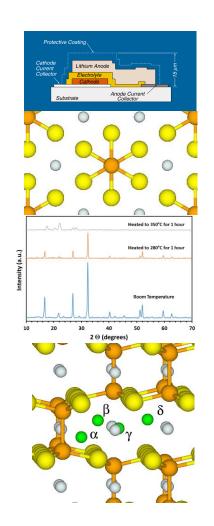


Structural analysis

> Thermal stability

Ionic conductivity

Summary and conclusions





**Motivation for studying Li<sub>4</sub>P<sub>2</sub>S<sub>6</sub>:** 

- Part of the search for the ideal solid electrolyte material for all-solid state Li ion batteries
- Li<sub>4</sub>P<sub>2</sub>S<sub>6</sub> frequently identified as unintended stable component of solid electrolyte preparations
- > Interesting structural properties, including disorder
- Transport and stability properties

#### **Previous work:**

JOURNAL OF SOLID STATE CHEMISTRY 43, 151-162 (1982)

## Synthese, structure cristalline et analyse vibrationnelle de l'hexathiohypodiphosphate de lithium Li<sub>4</sub>P<sub>2</sub>S<sub>6</sub>

R. MERCIER, J. P. MALUGANI, B. FAHYS, J. DOUGLADE,\* ET G. ROBERT

Laboratoire d'Electrochimie des Solides, ERA 810, et \*Laboratoire de Chimie Physique, Université de Franche-Comté, 25030 Besancon Cedex, France

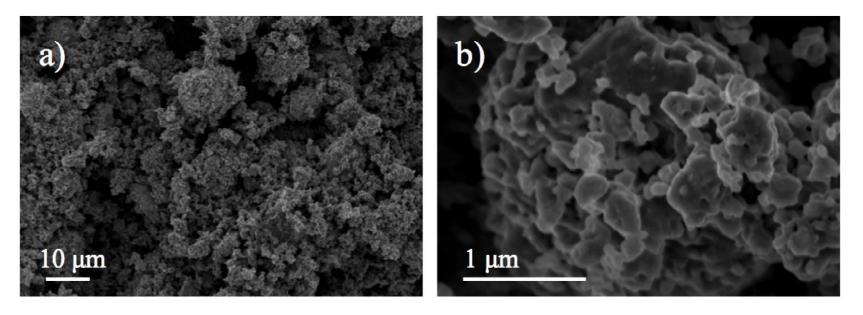


Synthesis:

$$2Li_2S + P_2S_5 \longrightarrow Li_4P_2S_6 + S$$

Sulfur removed by treatment with solvent; sample prepared for electrochemical applications using ball milling.

Scanning Electron Micrograph of prepared sample:



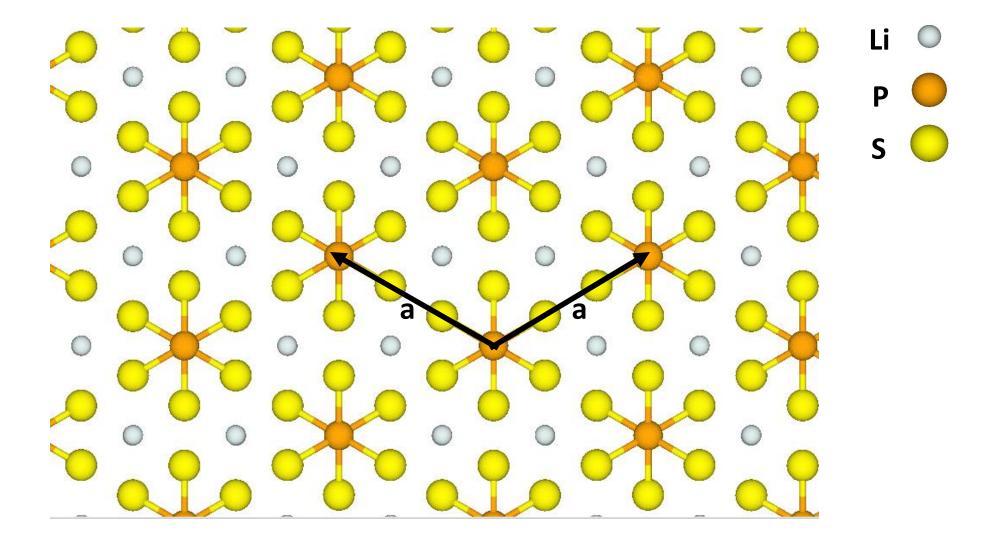


2015 APS March Meeting

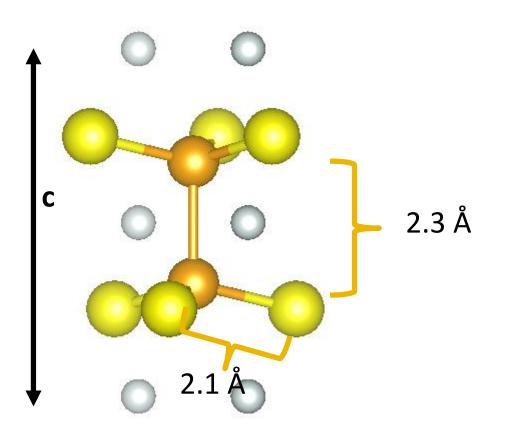
### **Computational methods**

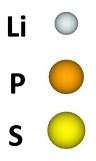
- Density functional theory with LDA
- PAW formalism using datasets generated with **ATOMPAW** code
- Electronic structure calculations performed using QUANTUM
  ESPRESSO and ABINIT codes
- Plane wave expansion for wave functions with  $|\mathbf{k} + \mathbf{G}|^2 \le 64 \text{ Ry}$
- Brillouin zone integration mesh of 0.003 bohr<sup>-3</sup>
- Visualization software: *Xcrysden* and *VESTA*; X-ray powder diffract simulated using *Mercury*

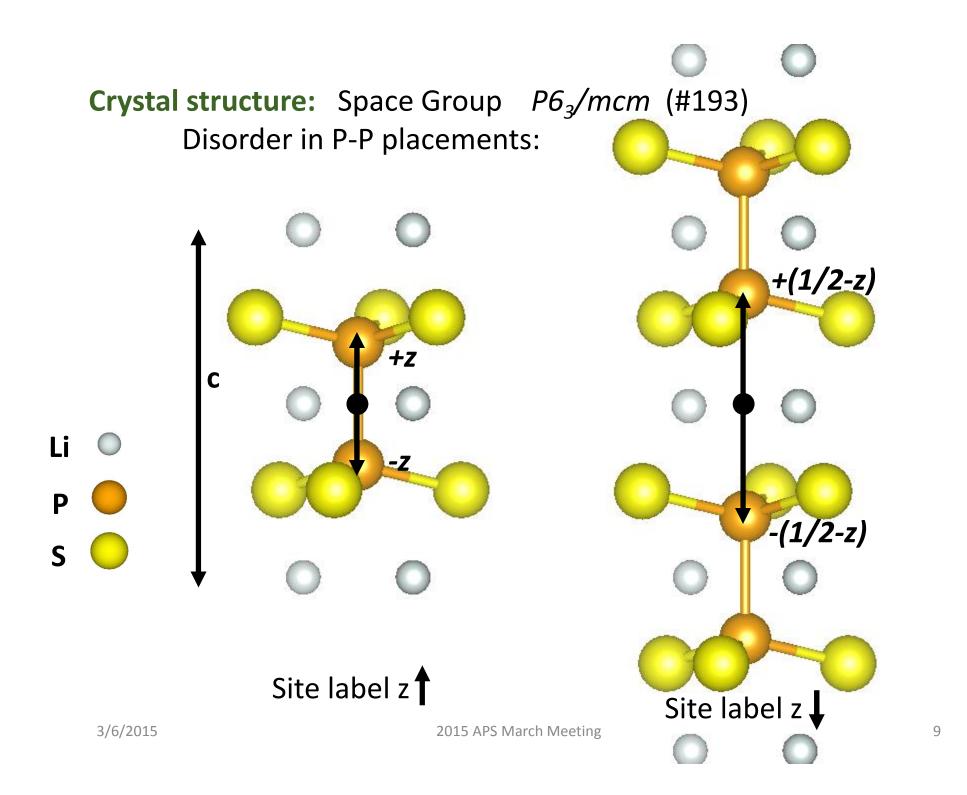
**Crystal structure:** Space Group  $P6_3/mcm$  (#193) Projection on to hexagonal plane:



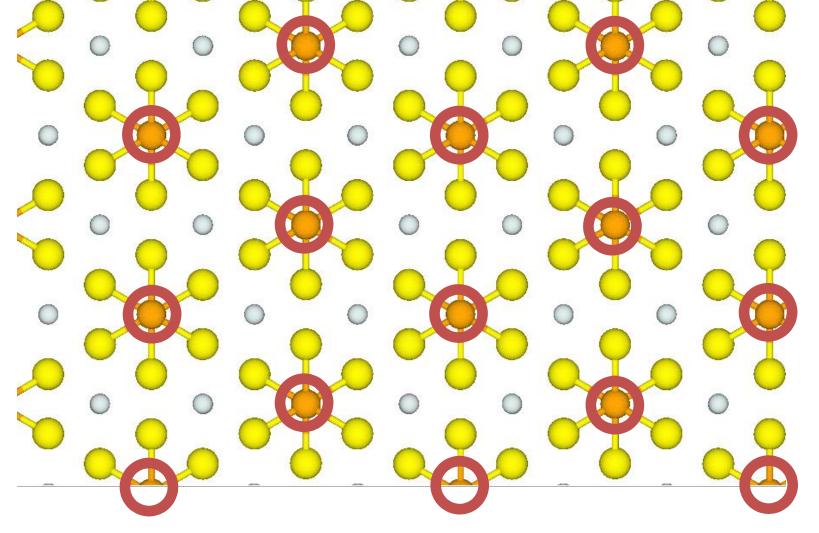
#### **Crystal structure:** Space Group $P6_3/mcm$ (#193) Li<sub>4</sub>P<sub>2</sub>S<sub>6</sub> units:







Structural variation can be mapped on to a two-dimensional hexagonal lattice with each P configuration taking  $\frac{1}{2}$  or  $\frac{1}{2}$  settings; Li and S configurations fixed



Li

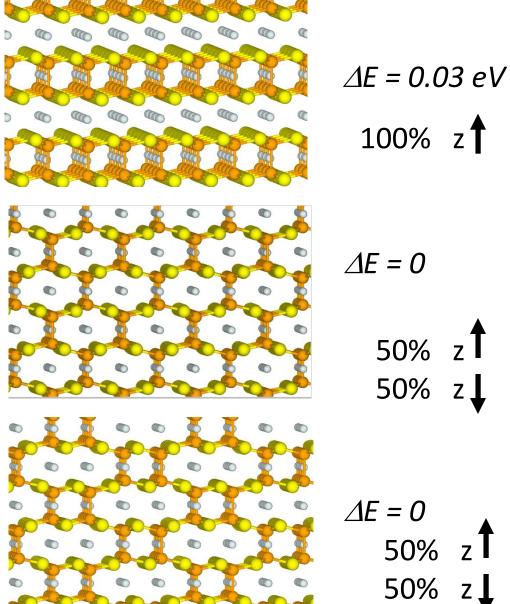
Ρ

S

 $\bigcirc$ 

#### **Examples:**

Structure "b"



100% z  $\Delta E = 0$ 50% z**T** 50% z Li  $\Delta E = 0$ Ρ

Structure "c"

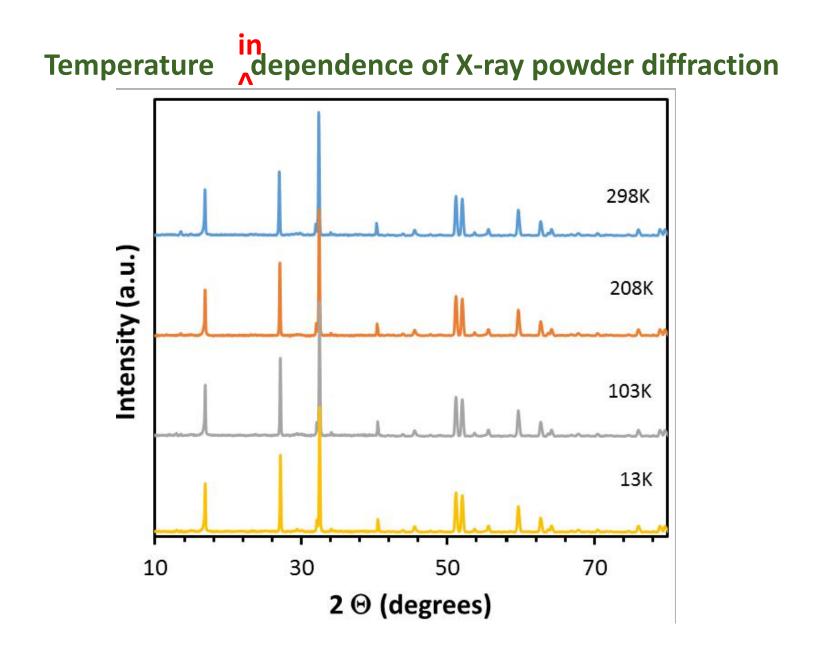
Structure "d"

2015 APS March Meeting

11

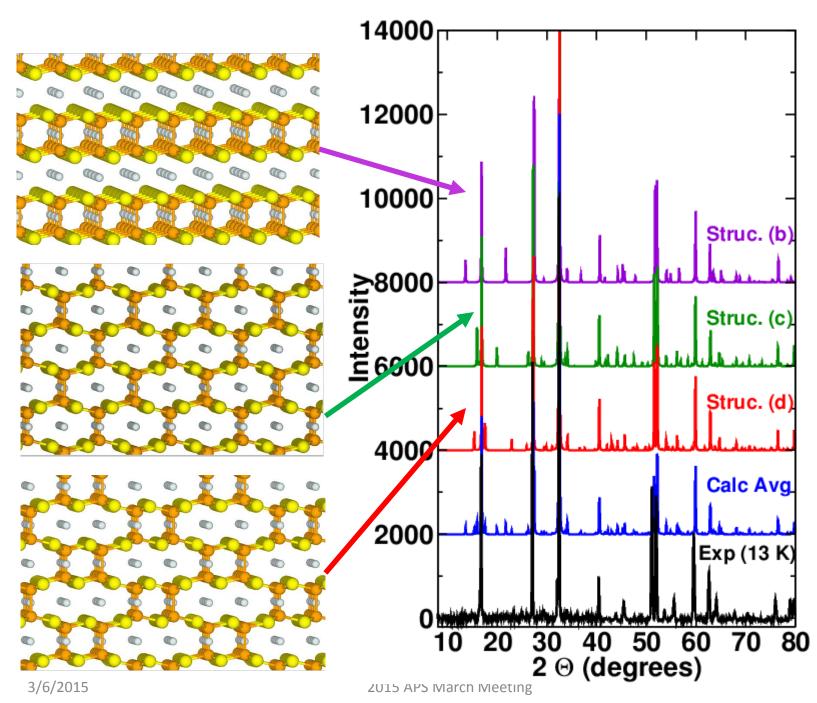
S

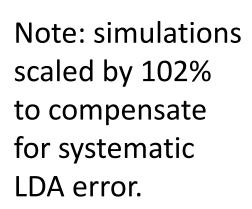
 $\bigcirc$ 



#### **Diffraction analysis**

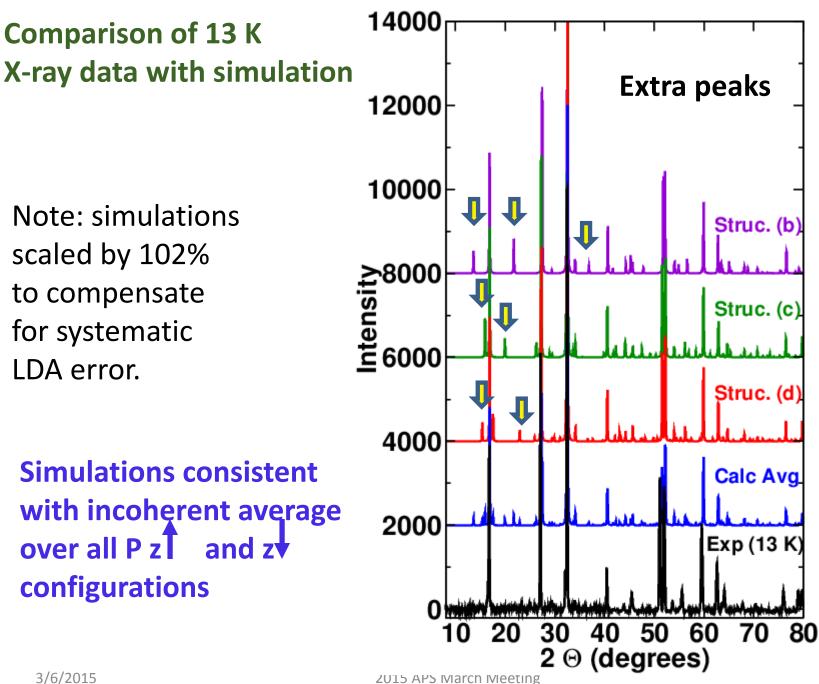
|                    | <i>a</i> (Å) | <i>c</i> (Å) | Z <sub>P</sub> | Z <sub>S</sub> |
|--------------------|--------------|--------------|----------------|----------------|
| Exp. 298K (X-ray)  | 6.0709       | 6.5903       | 0.1715         | 0.3237         |
| Exp. 13K (X-ray)   | 6.0747       | 6.5966       | 0.1715         | 0.3237         |
| Exp. 13K (neutron) | 6.0761       | 6.5961       | 0.1698         | 0.3284         |



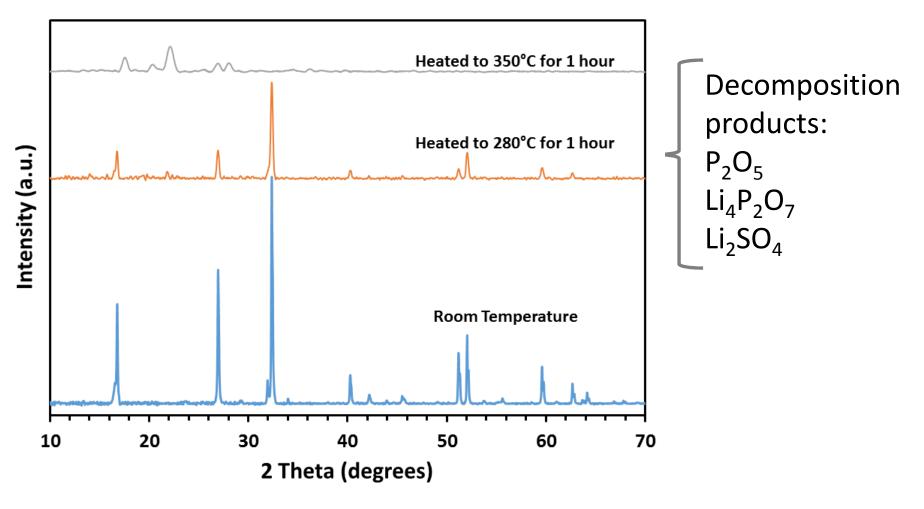


**Comparison of 13 K** 

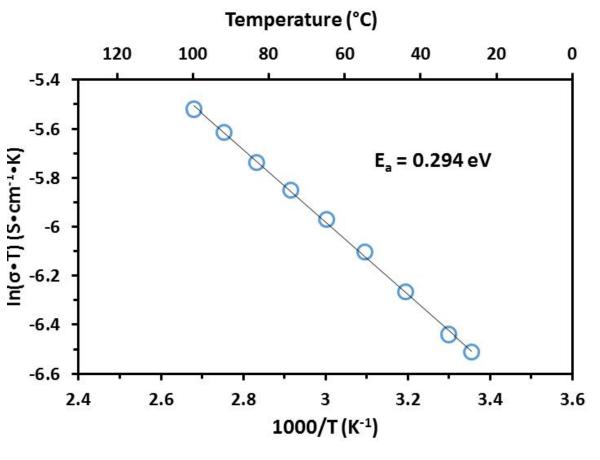
Simulations consistent with incoherent average over all P z and z configurations



**Stability:**  $Li_4P_2S_6$  is much less reactive than other lithium thio-phosphates, but it decomposes in air, especially at higher temperature

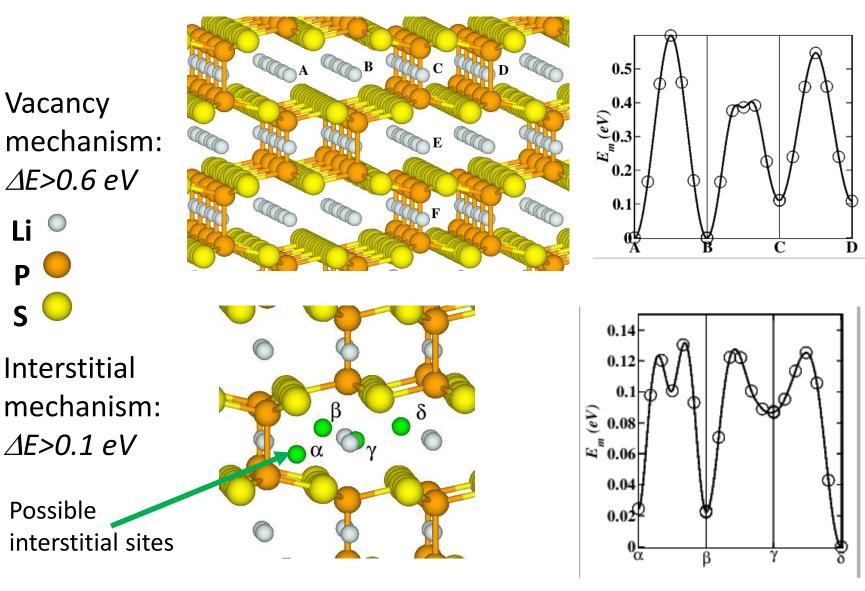


#### **Ionic conductivity and Activation Energy**



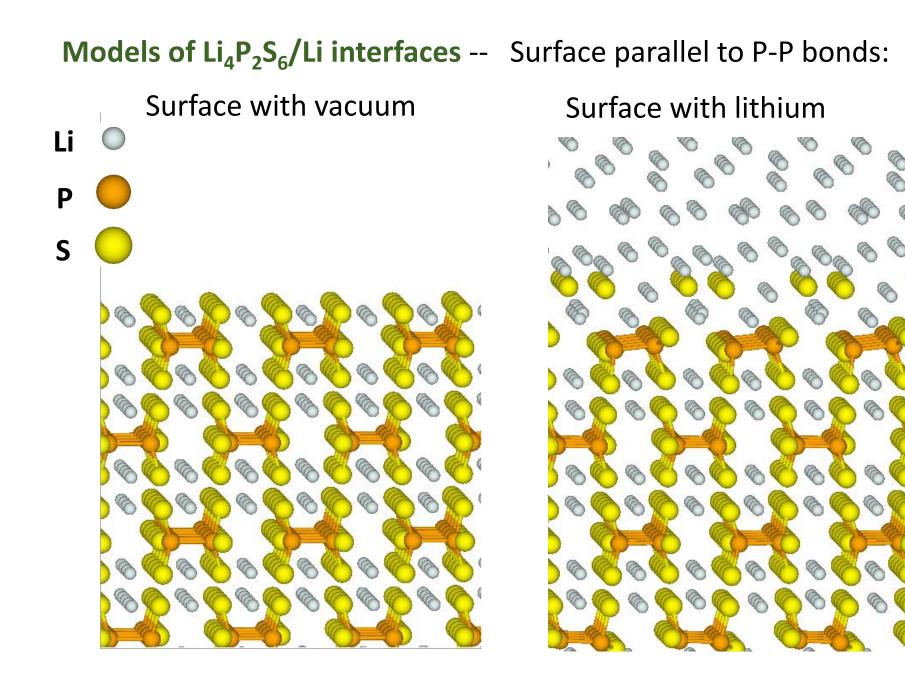
2.38 x  $10^{-7}$  S/cm at 25°C and 2.33 x  $10^{-6}$  S/cm at 100°C Li<sub>4</sub>P<sub>2</sub>S<sub>6</sub> pressed pellets with blocking (Al/C) electrodes

#### Simulations of ion mobility using Nudged Elastic Band Model

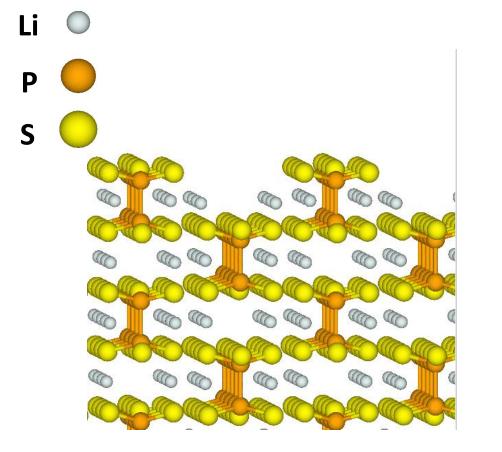


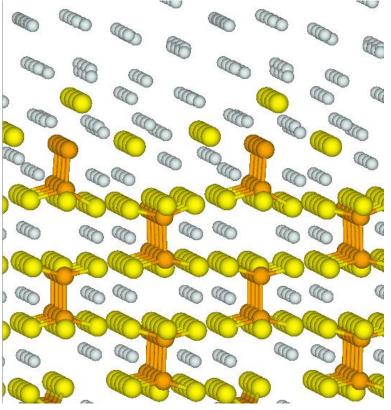
3/6/2015

S



## Models of Li<sub>4</sub>P<sub>2</sub>S<sub>6</sub>/Li interfaces --Surface perpendicular to P-P bonds:Surface with vacuumSurface with lithium





#### **Conclusions:**

- Diffraction results are consistent with Mercier's 1982 analysis with disorder on P sites due to small energy differences of alignment of P<sub>2</sub>S<sub>6</sub> fragments; remarkably temperature independent
- Small activation energy (E<sub>a</sub> = 0.3 eV) for ion conductivity consistent with interstitial mechanism
- > Thermal stability relative to other thio-phosphates
- Simulations of Li<sub>4</sub>P<sub>2</sub>S<sub>6</sub>/Li interfaces suggest that meta-stable buffer layers may be formed
- Further processing of materials needed to improve conductivity and stabilize Li/Li<sub>4</sub>P<sub>2</sub>S<sub>6</sub>/Li cells

