Understanding the phase properties of Li_2OHCl , $Li_2(OH)_{1-x}F_xCl$, and Li_2OHBr with first principles simulations

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Acknowledgements

This work was supported by NSF grant DMR-1507942.

Computations were performed on the Wake Forest University DEAC cluster, a centrally managed resource with support provided in part by the University.

Why study the phase properties of these materials

- The materials Li_2OHCl , $Li_2(OH)_{1-x}F_xCl$, and Li_2OHBr have been identified as possible electrolyte materials for all solid state Li-ion batteries. <u>Hood et al. Schwering et al. and Li et al.</u>
- The materials Li_2OHCl , $Li_2(OH)_{0.9}F_{0.1}Cl$ have been successfully cycled in electrochemical cells with metallic lithium electrodes. <u>Hood et al and Li et al.</u>
- The material Li₂OHCl experimentally known to exist two phases. Orthorhombic phase(~< 312k) and cubic phase (~>312K). The cubic phase has enhanced Li-ion conductivity.
- The material Li_2OHBr is known to exist in the cubic phase as low as 150K
- The material $Li_2(OH)_{0.9}F_{0.1}Cl$ is known to be stabilized in the cubic phase as low as room temperature.
- The question this work attempts to address is what is the physical differences between these materials that leads to the difference in phase properties.

1. Hood, Wang, Pandian, Keum, and Liang. J. Am. Chem. Soc. (2016), **138**, 1768–1771

3. Li, Zhou, Xin, Zhu, Lu, Cui, Jia, Zhou, Zhao, Goodenough. Angew. Chem. Int. Ed. (2016), 55, 9965-9968

^{2.} Schwering, Honnerscheid, Wullen, and Jansen. CHEMPHYSCHEM (2003), 4, 343 - 348

Structures

• All three materials are known to exist in a disordered cubic phase. <u>Schwering et al</u>





2/3 occupied lithium site

• The material Li_2OHCl is known to exist in an orthorhombic phase



Computational prediction for orthorhombic phase

Howard et al. Phys Rev Mat 1. 075406 2017

Why are $Li_2(OH)_{0.9}F_{0.1}Cl$, and Li_2OHBr stabilized in cubic structures compared to Li_2OHCl ?

- Why are Li_2OHBr and $Li_2(OH)_{0.9}F_{0.1}Cl$ stabilized cubic at room temperature? Why don't they form the orthorhombic structure?
- Because PV is small(~10^-5 eV) at atmospheric pressures use the Helmholtz free energy $F = \langle E \rangle -TS$
- This work uses Density functional theory calculations with the LDA within the projector augmented wave formalism(PAW) using the Quantum ESPRESSO code. ATOMPAW used to generate PAW datasets.

ATOMPAW code (Holzwarth et al. CPC 135, 329 (2001)) http://pwpaw.wfu.edu

QUANTUM ESPRESSO. (Giannozzi et al. JPCM 21, 394402 (2009); http://www.quantum-espresso.org, Gonze et al., CPC 180, 2582 (2009)); http://www.quantum-espresso.org, Gonze et al., CPC 180, 2582 (2009)); http://www.quantum-espresso.org, Gonze et al., CPC 180, 2582 (2009));

The Helmholtz Free energy approximation

- Helmholtz Free energy, $F = \langle E \rangle TS = -k_b T \log(Z)$. • $Z = \sum_{i=1}^{\Omega} e^{-\frac{U_i}{k_b T}} Z_{hp}^i = \sum_{i=1}^{\Omega} e^{-\frac{U_i + F_{hp}^i}{k_b T}}$
- This leads to the free energy $F_{tot} = \langle U_{SL} + F_{hp} \rangle TS_{\Omega}$,

•
$$< U_{SL} + F_{hp} > = \sum_{i=1}^{\Omega} (U_i + F_{hp}^i) \frac{e^{-\frac{U_i + F_{hp}^i}{k_b T}}}{Z}$$
, $S_{\Omega} = \frac{}{T} + k_b log(Z)$

Where U_{SL} is the "Static Lattice" energy calculated with ground state DFT And F_{hp}^{i} is the harmonic phonon free energy for configuration i $F_{hp}^{i} = kT \int_{0}^{\omega_{max}} \ln\left(\sinh\left(\frac{\hbar\omega}{2kT}\right)\right) g_{i}(\omega) d\omega$ $g_{i}(\omega)$ is the phonon density of states Fully disordered limit approximation of F(T) $S = \frac{\langle U_{SL} + F_{hp} \rangle}{T} + k_b \log(Z) \qquad T \to \infty \quad S = k_b \log(\Omega)$ $T \to \infty \langle U_{SL} + F_{hp} \rangle = \frac{1}{\Omega} \sum_{i=1}^{\Omega} (U_i + F_{hp}^i)$

If the OH rotations and lithium lattice are approximately completely coupled

 $\boldsymbol{\Omega}$ is determined by the combinatorics of this lithium lattice

$$\Omega = \frac{(3s^3)!}{(2s^3)!(s^3)!}$$
 where *s* is the cubic supercell dimension

Fully disordered (fd) approximation

$$F_{fd}(T) = \frac{1}{\Omega} \sum_{i=1}^{\Omega} (U_i + F_{hp}^i) - k_b T \log(\Omega) > F(T)$$



Comparisons of fully disordered Free energy of cubic and ortho phases



 $= \Delta F = \Delta U + \Delta F_{ph} - T\Delta S_{\Omega}$

For cubic phase 10 disordered models for ΔU And 2 for ΔF_{ph}

*Li*₂*OHCl*



Forth

Li₂OHBr



Disordered subpace approximation

•
$$F_{dss}(T) = \left[\frac{1}{m}\sum_{i=1}^{m} (U_i + F_{hp}^i) - k_b T\{ log(\Omega) + log\left(\frac{m}{\Omega}\right)\} \right]_{\min \to m}$$

•
$$F_{dss}(T) = S \rightarrow \infty \left\{ \left[\frac{1}{m'} \sum_{i=1}^{m'} (U_i + F_{hp}^i) - k_b T \{ log(\Omega) + log\left(\frac{m'}{s}\right) \} \right]_{\min \rightarrow m'} \right\}$$

• S is a collection of random samples of the configuration space(Ω) and m' is the number of lowest energy states kept from S

•
$$\frac{m'}{s} \approx \frac{m}{\Omega}$$

Conclusions

- Cubic phase of Li_2OHBr is predicted to be stabilized because of lower $\Delta U + \Delta F_{ph}$
- Experimentally known stabilization of the cubic phase of $Li_2(OH)_{1-x}F_xCl$ is predicted to be subtle effect. assuming ~ 0.02 eV error per formula unit. Suggest Orthorhombic phase will form within ~200-300K
- A method is laid out to calculate more accurate transition temperatures.

Future considerations

• Extending on the fully disordered approximation

•
$$F_{fd}(T) = \frac{1}{\Omega} \sum_{i=1}^{\Omega} (U_i + F_{hp}^i) - k_b T \log(\Omega)$$

- The partition function will effectively select some subset of the Ω congifurations
- Order the "energies" from least to greatest $U_1 + F_{hp}^1 \leq \dots \leq U_m + F_{hp}^m \leq \dots \leq U_\Omega + F_{hp}^\Omega$
- Imagine that a better model is that the system is a subpace (1 ... m) of the entire space $(1 ... \Omega)$
- Consider the "disordered subspace approximation" (dss)

$$F_{dss}(T) = \left[\frac{1}{m}\sum_{i=1}^{m} (U_i + F_{hp}^i) - k_b T\{ \log(\Omega) + \log\left(\frac{m}{\Omega}\right)\} \right]_{\min \to m} \approx F(T)$$

• $\frac{m}{\Omega}$ is the fraction of the entire space